

Keywords: *Mechanical Properties, Type 304L Stainless Steel, Type 316L Stainless Steel, Type 21-6-9 Stainless Steel, Hydrogen Embrittlement, J-Integral, Helium Embrittlement*

Retention: *Permanent*

2012 Accomplishments - Tritium Aging Studies on Stainless Steels

MICHAEL J. MORGAN
Materials Science and Technology

Publication Date: January 2013

This document was prepared in conjunction with work accomplished under Contract No. DE-AC09-08SR22470 with the U. S. Department of Energy

Savannah River National Laboratory
Savannah River Nuclear Solutions, LLC
Aiken, SC 29808

Prepared for the U.S. Department of Energy under
contract number DE-AC09-08SR22470.



DISCLAIMER

This work was prepared under an agreement with and funded by the U.S. Government. Neither the U.S. Government or its employees, nor any of its contractors, subcontractors or their employees, makes any expressed or implied:

1. Warranty or assumes any legal liability for the accuracy, completeness, or for the use or results of such use of any information, product, or process disclosed; or
2. Representation that such use or results of such use would not infringe privately owned rights; or
3. Endorsement or recommendation of any specifically identified commercial product, process, or service.

Any views and opinions of authors expressed in this work do not necessarily state or reflect those of the United States Government, or its contractors, or subcontractors.

Printed in the United States of America

**Prepared for
U.S. Department of Energy**

2012 Accomplishments - Tritium Aging Studies on Stainless Steels

CONTENTS	PAGE
List of Figures	ii
List of Tables	iii
I. Summary	1
II. Introduction	1
III. Fabrication And Tritium Charging Of Samples For Future Tritium Aging Studies	2
IV. Experimental Plan For Measuring Cracking Thresholds Of Tritium-Charged-And-Aged Steels In High Pressure Hydrogen Gas	7
V. Tritium Contents For Laboratory Inventory Requirements And Environmental Release Estimates	11
VI. Cracking Thresholds And Fracture Toughness Properties Of Tritium-Charged-And-Aged Stainless Steels	13
VII. The Effects Of Hydrogen, Tritium, And Heat Treatment On The Deformation And Fracture Toughness Properties Of Stainless Steels	16
VIII. Summary	19
IX. References	19

List of Figures	Page
Figure 1. Wedge Opening Load Specimen.	9
Figure 2. Wedge Opening Load Specimen (Detail).	10
Figure 3. Wedge-Opening-Load Specimen Showing Allen Wrench, Bolt for Loading the Specimen, and Grips for Unloading the Specimen on Mechanical Test Machine.	10
Figure 4. Wedge-Opening-Load Specimen Showing Bolt Loading Technique with Allen Wrench, COD Gage, and Leads For Crack Length Measurement.	11
Figure 5. Seventy-Five Arc-Shaped Stainless Steel Coupons (on left) Stacked onto Tritium Charging Assembly in Three Columns (on right).	12
Figure 6. Tritium Charging Assembly Showing Top and Bottom Caps and Six Rods.	12
Figure 7. Effect of Decay Helium Content on Sustained-Load Cracking Thresholds and Rising Load Fracture Toughness Values.	13
Figure 8. Figure 2. 2012 International Hydrogen Conference Presentation “Cracking Thresholds and Fracture Toughness Properties of Tritium-Charged-and-Aged Stainless Steels”.	15
Figure 9. Hydrogen-Precharged Steels Given a Prior Heat Treatment at 873 K Show Similar Fracture Toughness Reductions As Tritium-Precharged Samples.	17
Figure 10. Microstructures of (a) As Forged Steel (optical image); (b) Heat-Treated for 10 Min. at 873 K (Scanning Electron Microscope Image); and (c) Heat Treated for 10 Hours at 873 K (SEM Image).	17
Figure 11. 2012 International Hydrogen Conference Presentation on “The Effects of Hydrogen and Tritium and Heat Treatment on the Deformation and Fracture Toughness Properties of Stainless Steels.	18

List of Tables	Page
Table I - SRNL Data Sheet Charging Conditions Tritium Charging Run 2012-1.	3
Table II - Samples Charged During Tritium Charging Run 2012-1	4
Table III - SRNL Data Sheet Charging Conditions Tritium Charging Run 2012-2.	5
Table IV - Samples Charged During Tritium Charging Run 2012-2	6

2012 Accomplishments - Tritium Aging Studies on Stainless Steels

I. SUMMARY

This report summarizes the research and development accomplishments during FY12 for the tritium effects on materials program. The tritium effects on materials program is designed to measure the long-term effects of tritium and its radioactive decay product, helium-3, on the structural properties of forged stainless steels which are used as the materials of construction for tritium reservoirs. The FY12 R&D accomplishments include: (1) Fabricated and Thermally-Charged 150 Forged Stainless Steel Samples with Tritium for Future Aging Studies; (2) Developed an Experimental Plan for Measuring Cracking Thresholds of Tritium-Charged-and-Aged Steels in High Pressure Hydrogen Gas; (3) Calculated Sample Tritium Contents For Laboratory Inventory Requirements and Environmental Release Estimates; (4) Published report on “Cracking Thresholds and Fracture Toughness Properties of Tritium-Charged-and-Aged Stainless Steels”; and, (5) Published report on “The Effects of Hydrogen, Tritium, and Heat Treatment on the Deformation and Fracture Toughness Properties of Stainless Steels”. These accomplishments are highlighted here and references given to additional reports for more detailed information.

II. INTRODUCTION

Forged stainless steels are used as the materials of construction for tritium reservoirs. During service, tritium diffuses into the reservoir walls and radioactively decays to helium-3. Tritium and decay helium cause a higher propensity for cracking which could lead to a tritium leak or delayed failure of a tritium reservoir. The factors that affect the tendency for crack formation and propagation include: (a) time of exposure; (b) steel type; (c) steel microstructure; (d) reservoir geometry and gas pressure; and, (e) reservoir residual stresses from welding and manufacturing. Fracture toughness properties are needed for designing tritium reservoirs and evaluating the long-term effects of tritium on their structural properties. These effects are being characterized with the Enhanced Surveillance Campaign Tritium Effects on Materials Program using the plan described in Reference 1. Chiefly, the results are obtained by measuring the effects of tritium on the tensile and fracture toughness properties of samples fabricated from forgings on pre-charged samples tested in air.

This report describes the FY12 accomplishments for the Tritium Effects on Materials Program. The FY12 R&D accomplishments include: (1) Fabricated and Thermally-Charged 150 Forged Stainless Steel Samples with Tritium for Future Aging Studies; (2) Developed an Experimental Plan for Measuring Cracking Thresholds of Tritium-Charged-and-Aged Steels in High Pressure Hydrogen Gas; (3) Calculated Sample Tritium Contents For Laboratory Inventory Requirements and Environmental Release Estimates; (4) Published report on “Cracking Thresholds and Fracture Toughness Properties of Tritium-Charged-and-Aged Stainless Steels”; and, (5) Published report on

“The Effects of Hydrogen, Tritium, and Heat Treatment on the Deformation and Fracture Toughness Properties of Stainless Steels”.

III. FABRICATION AND TRITIUM CHARGING OF SAMPLES FOR FUTURE TRITIUM AGING STUDIES

An important FY12 accomplishment was the fabrication and tritium precharging of 150 samples for future tritium aging studies. These samples are part of four new comprehensive experimental research and development programs that are underway for investigating the effects of hydrogen, tritium, and decay helium on the fracture toughness properties of forged stainless steels. The programs and test matrices are described in a recent program plan (1).

These programs are first-of-a-kind because they set out to measure tritium and decay helium effects on the cracking properties of stainless steels using actual tritium reservoir forgings instead of the experimental forgings of past programs. In this way, the properties measured will be nearly identical to actual reservoir properties because the microstructure of the samples will be like that of the forged reservoirs. There are four major programs each designed to measure the effects of a specific forging variable on tritium compatibility.

The programs include three stainless steels, multiple yield strengths, four different forging processes, and four different reservoir forgings. They are entitled: (1) Orientation Effect - Type 316L Cup Forging; (2) Yield Strength Effect - Type 304L Cylindrical Block Forging; (3) Fracture Toughness Variability - Type 21-6-9 Stainless Brick Forging; and (4) Forging Process Effects. The plan describes the goals of each program, lists the sample test matrices and experimental conditions, and reports on initial results (1).

Two tritium charging runs were conducted in FY12 (2-3) by Tritium Programs Operations and Engineering with support by SRNL R&D Engineering. The tritium fill conditions were designed to saturate the samples with tritium at a nominal pressure of 5000 psia at a temperature of 350 C for two weeks. Table I shows the SRNL Data Sheet for the specific fill conditions that were used to charge the samples in the first run and the samples charged during the run are listed in Table II (2). Likewise, Table III shows the SRNL Data Sheet for the fill conditions that were used during the second run and the samples charged are listed in Table IV (3).

Table I SRNL DATA SHEET**CHARGING CONDITIONS TRITIUM CHARGING RUN 2012-1****FORGING EFFECT SAMPLES**

Document Number: SRNL-L4400-2012-00027 Rev 0

Vessel Serial Number: 02151151-4

Vessel Charge History: First Use

Charge Type: Type 1 (C-Shaped Samples)

Total Free Volume, Vessel and Tubing: 248 cc

Acceptable Temperature Range During Fill: 18-28°C

Target Fill Pressure: 2619 psia

Target Fill Pressure Range: 2609-2629 psia

Heated Vessel Temperature: 350°C ± 5°C

Duration at Temperature and Pressure: 14 Days ± 8 Hours

Table II Samples Charged During Tritium Charging Run 2012-1**VESSEL SN 02151151-4****FORGING EFFECT SAMPLES – Stacking Order**

Source	Test Env.	Material	ID	ID	ID	Layer in Rack
No Anneal Herf	Air	304L	F16-1	F16-5	F16-9	25
No Anneal Herf	Air		F71-1	F71-5	F71-9	24
No Anneal Mech	Air	304L	M16-1	M16-5	M16-9	23
No Anneal Mech	Air		M71-1	M71-6	M71-9	22
No Anneal Hyd.	Air	304L	Y16-1	Y16-5	Y16-9	21
No Anneal Hyd.	Air		Y71-1	Y71-5	Y71-9	20
No Anneal Screw	Air	304L	S16-1	S16-5	S16-9	19
No Anneal Screw	Air		S71-2	S71-5	S71-9	18
Stage 2	Air	304L	S2A-4	S2B-3	S2B-5	17

Sandia Forging Samples

Stem	Air	316L	26AL6	26AL9	26BL5	16
	Air		26AL5	26AL10	26BL6	15
	Air		26AL4	26BL1	26BL7	14
	Air		26AL3	26BL2	26BL8	13
Cup	Air	316L	26RE3	26RF5	26RC2	12
	Air		26RD4	26RE1	26RF3	11
	Air		26RC5	26RD2	26RE4	10
	Air		26RF1	26RC3	26RD5	9
Stem	H2	316L	26AL13	26AL19	26BL15	8
	H2		26AL14	26AL20	26BL16	7
	H2		26AL15	26BL11	26BL17	6
	H2		26AL16	26BL12	26BL18	5
Cup	H2	316L	26RE8	26RF10	26RC7	4
	H2		26RD9	26RE6	26RF8	3
	H2		26RC10	26RD7	26RE9	2
	H2		26RF6	26RC8	26RD10	1

Rack Info.

Bottom = 1, Top = 25

Total Samples: 75

**Table III SRNL DATA SHEET
CHARGING CONDITIONS TRITIUM CHARGING RUN 2012-2**

BLOCK FORGING SAMPLES

Document Number: SRNL-L4400-2012-00027 Rev 0

Vessel Serial Number: 02151151-3

Vessel Charge History: First Use

Charge Type: Type 1 (C-Shaped Samples)

Total Free Volume, Vessel and Tubing: 253 cc

Acceptable Temperature Range During Fill: 18-28°C

Target Fill Pressure: 2619 psia

Target Fill Pressure Range: 2609-2629 psia

Heated Vessel Temperature: 350°C ± 5°C

Duration at Temperature and Pressure: 14 Days ± 8 Hours

Table IV Samples Charged During Run 2012-2**VESSEL SN 02151151-3****FORGING EFFECT SAMPLES – Stacking Order****KC AND SANDIA SAMPLES**

Source	Test Env.	Material	ID	ID	ID	Layer in Rack
KCP Remnants						
Anneal Herf	Air	304L	AF16-3	AF16-5	AF16-7	25
Anneal Herf	Air		AF71-3	AF71-5	AF71-7	24
Anneal Mech	Air	304L	AM16-3	AM16-4	AM16-8	23
Anneal Mech	Air		AM71-3	AM71-4	AM71-8	22
Anneal Hyd.	Air	304L	AY16-3	AY16-5	AY16-7	21
Anneal Hyd.	Air		AY71-3	AY71-4	AY71-8	20
Anneal Screw	Air	304L	AS16-3	AS16-4	AS16-8	19
Anneal Screw	Air		AS71-3	AS71-4	AS71-8	18
Stage 2	Air	304L	S2B-1	S2B-6	S2A-2	17
Sandia Forging Samples						
11459 (LY)	Air	304L	59RC7	59RD11	59RA5	16
	Air		59RB10	59RC3	59RD8	15
	Air		59RA2	59RB7	59RC12	14
	Air		59RD5	59RA10	59RB3	13
11460 (HY)	Air	304L	60RC7	60RD11	60RA5	12
	Air		60RB10	60RC3	60RD3	11
	Air		60RA2	60RB7	60RC12	10
	Air		60RD5	60RA10	60RB3	9
11459 (LY)	H2	304L	59RA3	59RB8	59RC1	8
	H2		59RD6	59RA11	59RB4	7
	H2		59RC9	59RD2	59RA7	6
	H2		59RB12	59RC5	59RD10	5
11460 (HY)	H2	304L	60RA3	60RB8	60RC1	4
	H2		60RD6	60RA11	60RB4	3
	H2		60RC9	60RD2	60RA7	2
	H2		60RB12	60RC5	60RD10	1

Rack Info. Bottom = 1, Top = 25

Note: AF16-4, AF71-4 and AY16-4 were changed to -5 because precracks were too long.

Note: AF16-8, AF71-8 and AY16-8 were changed to -7 because precracks were too long.

Total samples: 75

IV. EXPERIMENTAL PLAN FOR MEASURING CRACKING THRESHOLDS OF TRITIUM-CHARGED-AND-AGED STEELS IN HIGH PRESSURE HYDROGEN GAS

Prior experiments and analysis have demonstrated that sustained-load cracking threshold data are needed to conduct fracture mechanics analyses and establish safe operating lifetimes of tritium reservoirs (4-6). Tritium and decay helium reduce the fracture toughness properties of stainless steel and make crack nucleation and propagation easier. Until now, these effects have been characterized in the laboratory by measuring the fracture toughness properties as a function of tritium and decay helium content on samples tested in air. Recent results suggest that stainless steels tested in high-pressure hydrogen environments could have lower cracking thresholds than steels pre-charged with hydrogen and tested in air. A similar result is expected for samples tested in high-pressure tritium environments when compared to tritium pre-charged steels, however, a facility for conducting fracture mechanics tests on radioactive tritium samples is not available at this time.

An experimental concept for acquiring fracture mechanics properties on tritium pre-charged steels tested in high-pressure hydrogen environments under sustained loads was developed during FY12. The concept is modeled after ASTM E1681 "Standard Test Method for Determining Threshold Stress Intensity Factor for Environmental Assisted Cracking of Metallic Materials" (7) and utilizes bolt-loaded samples and an existing tritium pre-charging facility at SRS. This concept could provide needed sustained-load fracture mechanics data in the short term until a mechanical testing facility is available for conducting rising-load mechanical property and fracture mechanics tests on tritium-charged-and-aged samples tested in high-pressure hydrogen environments (8).

Cracking threshold experiments on tritium-precharged samples in high-pressure deuterium gas would be conducted in the following way. A wedge-opening load (WOL) compact tension specimen as shown in Figures 1 and 2 will be used for the tests. Samples will be fabricated from stainless steel brick forgings (1) with a width of 1 inch and a thickness of .25 inches. The samples will be fatigue pre-cracked and then loaded into a tritium charging vessel like those used for charging the samples in Section III above.

The vessel will then be sealed and pressurized on the tritium loading line in the same way samples have been tritium pre-charged for the prior studies; i.e., the vessel will be pressurized to 5000 psi and held at a temperature of 350°C for up to four weeks to saturate the samples with tritium. A two-week longer charge time will be needed to reach saturation because these samples are 0.05" thicker than previous samples.

After the tritium pre-charging operation, the vessel will be cooled down and evacuated. The samples will be removed from the vessel and transported to SRNL for freezer storage at -70 C to minimize tritium off-gassing losses. The samples will be stored and aged for up to five years to build-in helium from tritium. After a set of samples reach the desired helium content from tritium decay, they will be loaded with a bolt to specific crack-opening displacement (COD) level (Figures 3 and 4). The COD will

correspond to an initial load derived from the load-COD calibration curve. A potential-drop system may also be used to monitor crack length during the loading operation.

Identical samples with the desired helium content will be loaded to low, medium, and high levels of COD. These bolt-loaded samples will then be placed back in a tritium charging vessel on the tritium loading line. Each vessel will be able to accommodate about 20 bolt-loaded specimens. The vessel will then be back-filled with high-pressure hydrogen or deuterium gas and then valved off and held at ambient temperature in the loading line glove box for up to three months at pressure. This hold step will provide a high-pressure gas environment around the loaded crack tip in a tritium-precharged sample with a given helium content.

After three months, the vessel will be evacuated and the samples recovered and transferred to SRNL. The samples will be heat-tinted in a furnace to mark the extent of any crack growth, and unloaded in a mechanical testing machine while measuring the current load on the bolt. Again, a potential drop system may be used to measure any crack growth that may have occurred during the high-pressure exposure. The load on the bolt and the measured crack length at the end of the hold period will be used to establish the cracking threshold at the current helium level in the sample.

For samples that cracked during the hold period, new samples will be loaded using low, medium, and high loads, but now lower than the values that caused cracking in the earlier experiment. In this way, a more precise cracking threshold value will be found. Conversely, if samples do not crack during the defined hold period, a safe operating load is established. Subsequent samples will be loaded at higher loads to establish the cracking threshold.

More detailed procedures will be developed during FY13 using hydrogen exposures on non-charged samples.

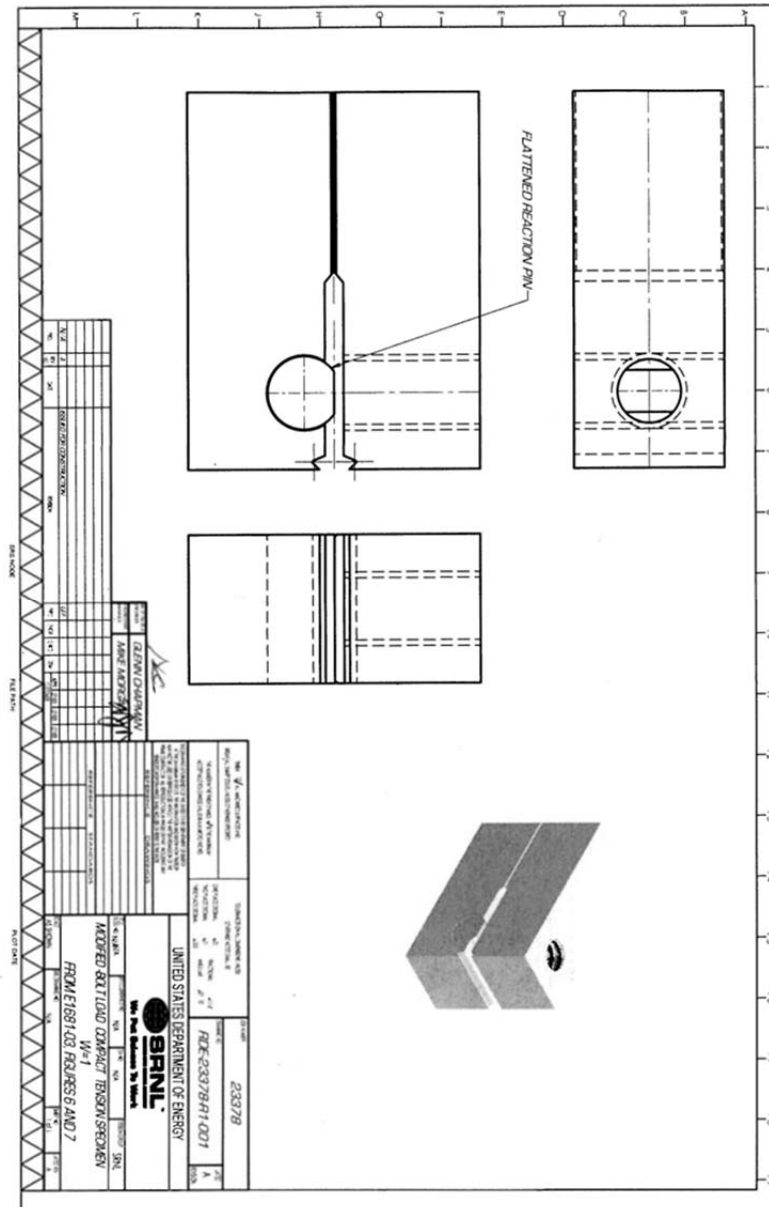


Figure 1. Wedge-Opening-Load Specimen.

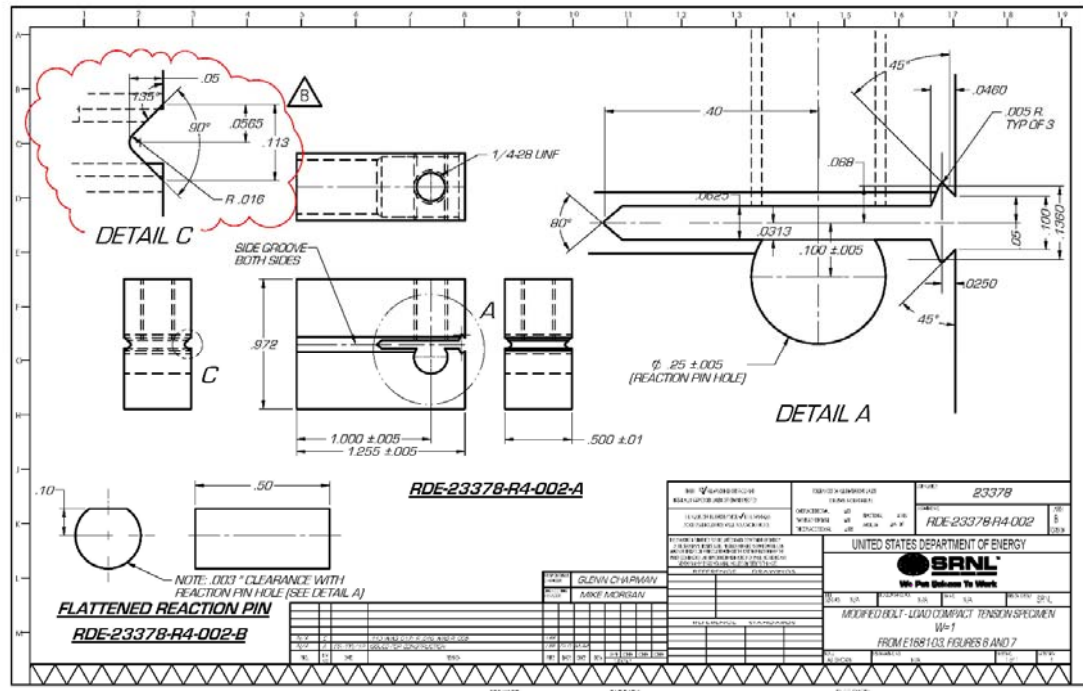


Figure 2. Wedge Opening Load Specimen (Detail).



Figure 3. Wedge-Opening-Load Specimen Showing Allen Wrench, Bolt for Loading the Specimen, and Grips for Unloading the Specimen on Mechanical Test Machine.

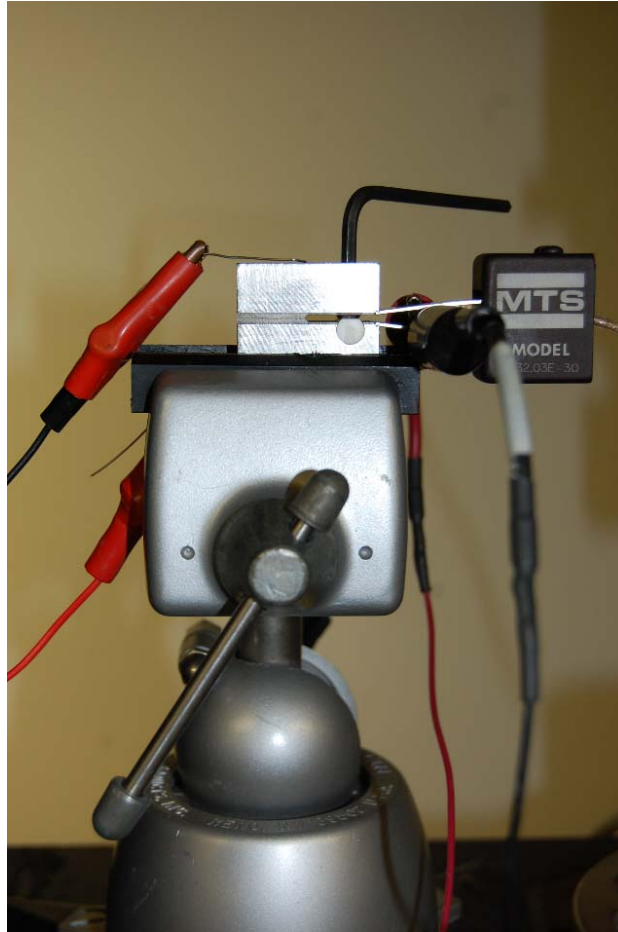


Figure 4. Wedge-Opening-Load Specimen Showing Bolt Loading Technique with Allen Wrench, COD Gage, and Leads For Crack Length Measurement.

V. TRITIUM CONTENTS AND FOR LABORATORY INVENTORY REQUIREMENTS AND ENVIRONMENTAL RELEASE ESTIMATES

Two tritium charging runs were conducted for SRNL during October of 2012 and November of 2012. The samples are listed in Section III above. In order to satisfy SRNL tritium inventory requirements and to provide future estimates of tritium release during testing the total tritium dissolved in the samples after charging, a tritium diffusion calculation was conducted (9-10). The calculations were performed using the Diff computer program (11). The Diff computer program will also be used for estimating tritium off-gassing rates once the samples are transferred to SRNL and tested and a more complete exposure history is defined. For the off-gassing calculation, the procedure described in Reference 12 will be used.

There were seventy-five arc-shaped stainless steel coupons stacked together for each charging assembly for the two tritium charging runs conducted for SRNL on the Tritium Loading Line during October and November, 2012 (Figures 5-6). The samples were stacked on a stainless steel assembly shown in Figure 6. The assembly consists of two stainless steel caps 1.94" in diameter by .170" thick and six .170" diameter x 5.5" long rods and a .75" long x .5" wide x .170" thick handle. The tritium charging was conducted at temperature of 350°C and a fill pressure at temperature of approximately 5000 psia.

Calculations were performed that resulted in the following findings (9-10) that will be used for inventory control of the samples stored in the laboratory: For the first charging run, the total tritium content for the samples and charging assembly was calculated to be 1908 Curies. The seventy-five samples have a tritium content of 19.3 Curies each for a total of 1449 Curies. The samples are stacked on a stainless steel assembly consisting of two caps, 6 rods and a handle. The stainless steel assembly has a tritium content of 459 Curies. For the second tritium charging run conducted for SRNL during November of 2012. The total tritium content for the samples and charging assembly was calculated to be 1888 Curies. The seventy-five samples have a tritium content of 19.1 Curies each for a total of 1432 Curies. In this case, the stainless steel assembly has a tritium content of 456 Curies.

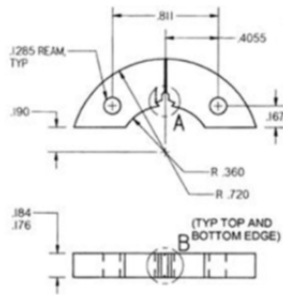


Figure 5. Seventy-Five Arc-Shaped Stainless Steel Coupons (on left) Stacked onto Tritium Charging Assembly in Three Columns (on right).



Figure 6. Tritium Charging Assembly Show Top and Bottom Caps and Six Rods

VI. CRACKING THRESHOLDS AND FRACTURE TOUGHNESS PROPERTIES OF TRITIUM-CHARGED-AND-AGED STAINLESS STEELS

A report entitled “Cracking Thresholds and Fracture Toughness Properties of Tritium-Charged-and-Aged Stainless Steels” was presented at the 2012 International Hydrogen Conference and submitted for publication in the conference proceedings (13-14). Cracking thresholds and fracture toughness properties were measured for hydrogen and tritium pre-charged Types 304L and 21-6-9 stainless steels. The purpose of the experiments was to measure the effect of decay helium on the fracture properties of stainless steels and to compare sustained-load cracking thresholds with rising-load fracture toughness values. Sustained-load cracking threshold tests were conducted by step-loading and holding tritium pre-charged samples at constant loads until crack extension was detected. Rising-load fracture toughness values were measured using ASTM E1820. The results show that while both cracking thresholds and fracture toughness values decreased with increasing ^3He content (Fig. 7), cracking thresholds were lower.

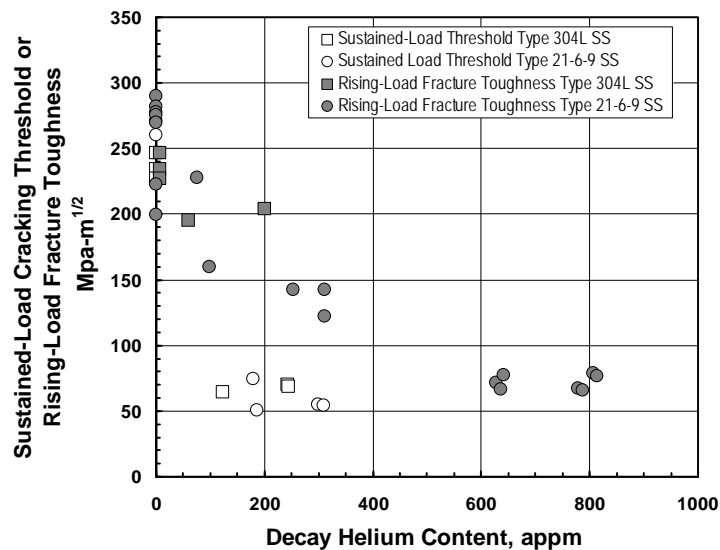


Figure 7. Effect of Decay Helium Content on Sustained-Load Cracking Thresholds and Rising Load Fracture Toughness Values.

The results of this study have important implications for tritium storage vessels. First, the results indicate that the threshold for tritium-induced cracking depends on the age of a vessel and its exposure history to tritium. During service, the vessel will age and become more embrittled with time because of tritium dissolution, diffusion, and radioactive decay. The effect of tritium and decay helium on cracking threshold and fracture

toughness (Fig. 7) means that tritium vessels may become susceptible to subcritical cracking and delayed failure. Cracking threshold values like those shown in Figure 7 are needed for establishing safe lifetimes for vessels in tritium service. In this study, sustained-load cracking thresholds were lower than rising-load cracking thresholds.

The results suggest that the difference between sustained-load cracking thresholds and rising-load cracking thresholds is not due to problems associated with detecting the actual point of crack extension. Sustained-load thresholds are still lower than rising-load thresholds even after adjustments are made to the rising-load data to account for the actual point of crack extension. Rather, the results suggest that sustained-load cracking thresholds are lower because of greater tritium diffusion and redistribution during the cracking process. Sustained-load tests apparently have sufficiently longer loading times for greater tritium redistribution and concentration near the crack tip. The higher crack tip tritium concentration causes a lower cracking threshold.

Because of the uncertainties associated with the establishing hold times that are sufficiently long enough for the tritium-induced cracking process, the sustained-load testing protocol needs further work. It may be possible to refine the test protocol using hydrogen pre-charged samples so that fewer samples would be needed for the tritium tests. Alternatively, a correlation between long-time sustained-load tests results and the short-term rising-load test results could be developed. These will be subjects of future investigations.

The complete presentation is depicted in Figure 8.

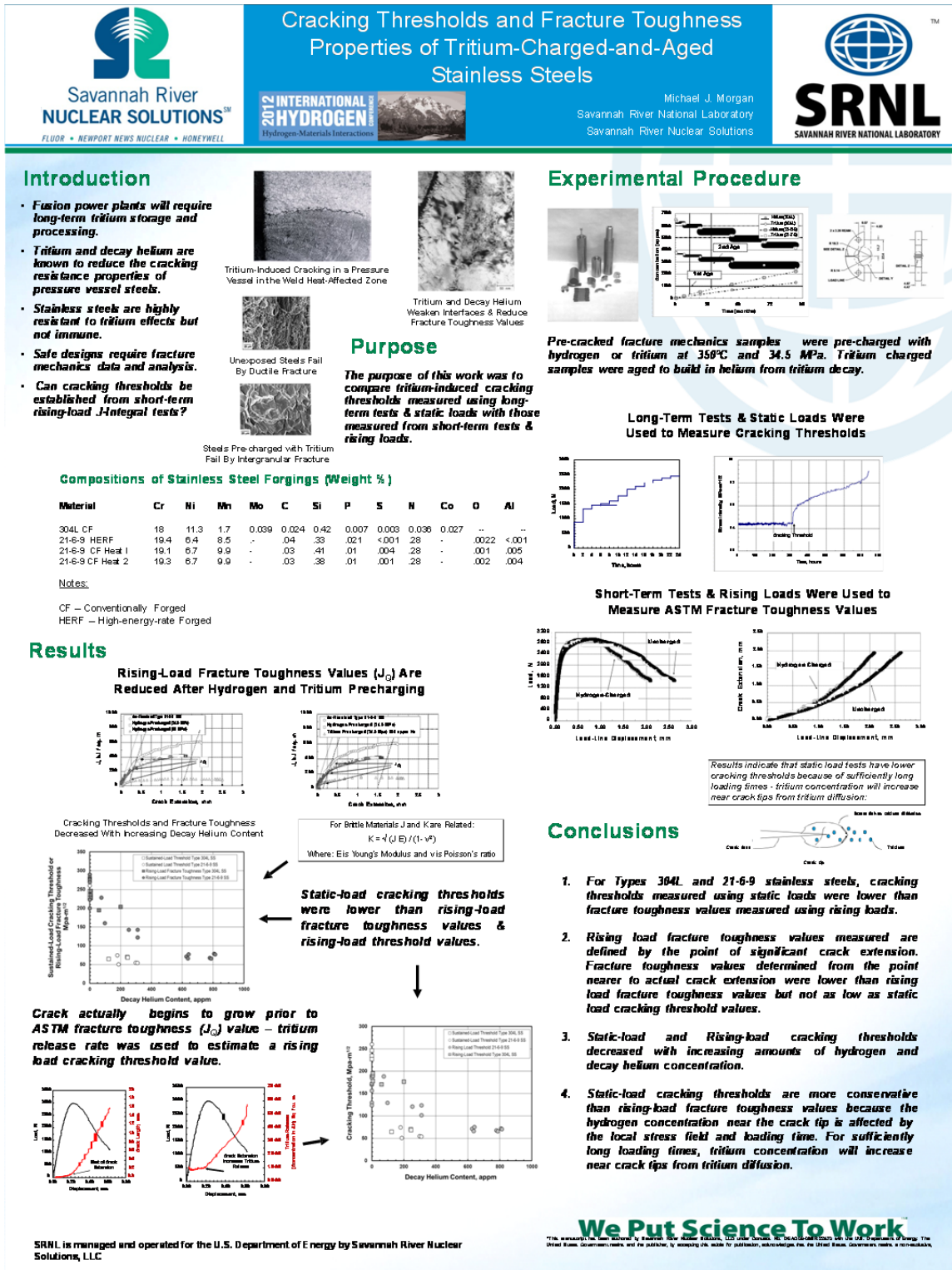


Figure 8. 2012 International Hydrogen Conference Presentation “Cracking Thresholds and Fracture Toughness Properties of Tritium-Charged-and-Aged Stainless Steels”.

VII. THE EFFECTS OF HYDROGEN, TRITIUM, AND HEAT TREATMENT ON THE DEFORMATION AND FRACTURE TOUGHNESS PROPERTIES OF STAINLESS STEELS

A report on the effects of hydrogen, tritium, and heat treatment on the deformation and fracture toughness properties of stainless steels was also presented at the 2012 International Hydrogen Conference and submitted for publication in the conference proceedings (15-16). In this work, the deformation and fracture toughness properties of forged stainless steels pre-charged with tritium were compared to the deformation and fracture toughness properties of the same steels heat treated at 773 K or 873 K and pre-charged with hydrogen (Fig. 9). Forged stainless steels pre-charged with tritium exhibit an aging effect: Fracture toughness values decrease with aging time after pre-charging because of the increase in concentration of helium from tritium decay. The study showed that forged stainless steels given a prior heat treatment and then pre-charged with hydrogen also exhibit an aging effect: Fracture toughness values decrease with increasing time at temperature. A microstructural analysis showed that the fracture toughness reduction in the heat-treated steels was due to patches of recrystallized grains that form within the forged matrix during the heat treatment (Fig. 10). The combination of hydrogen and the patches of recrystallized grains resulted in more deformation twinning. Heavy deformation twinning on multiple slip planes was typical for the hydrogen-charged samples; whereas, in the non-charged samples, less twinning was observed and was generally limited to one slip plane. Similar effects occur in tritium pre-charged steels, but the deformation twinning is brought on by the hardening associated with decay helium bubbles in the microstructure. The complete presentation is depicted in Figure 11.

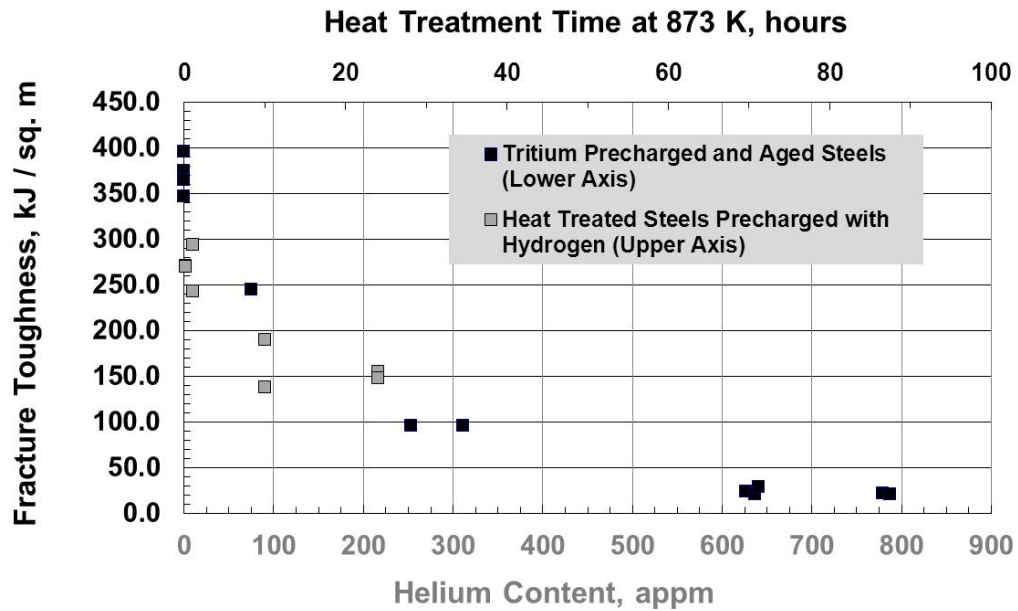


Figure 9. Hydrogen-Precharged Steels Given a Prior Heat Treatment at 873 K Show Similar Fracture Toughness Reductions As Tritium-Precharged Samples.

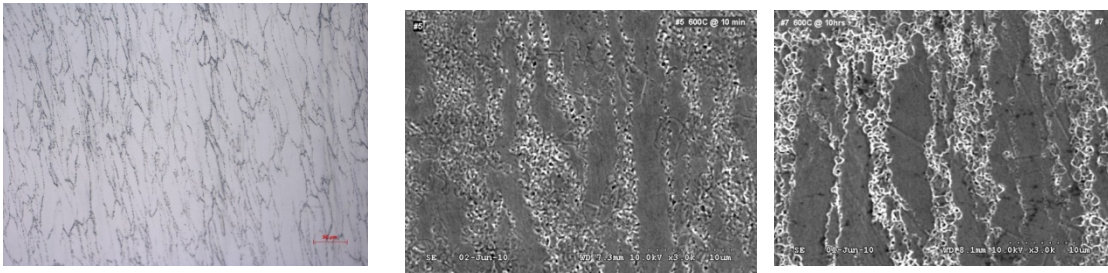


Figure 10. Microstructures of (a) As Forged Steel (optical image); (b) Heat-Treated for 10 Min. at 873 K (Scanning Electron Microscope Image); and (c) Heat Treated for 10 Hours at 873 K (SEM Image).

THE EFFECTS OF HYDROGEN, TRITIUM, AND HEAT TREATMENT ON THE DEFORMATION AND FRACTURE TOUGHNESS PROPERTIES OF STAINLESS STEEL

2012 INTERNATIONAL HYDROGEN



M.J. MORGAN, M. H. TOSTEN, AND G.K. CHAPMAN
Savannah River National Laboratory
Savannah River Nuclear Solutions

Introduction

- Tritium gas is processed and stored at the Savannah River Site in stainless steel vessels.
- Austenitic stainless steels are highly compatible with tritium but, after long term use, tritium and its decay helium reduce fracture toughness and can induce subcritical crack growth.
- The fracture toughness reduction over time is a form of hydrogen embrittlement that is enhanced by the effects of helium on the deformation properties.
- Heat-treated stainless steels also exhibit an aging effect after they are pre-charged with hydrogen. Fracture toughness values decrease with increasing time at temperature in the sensitization temperature range, 773-873 K.
- Are the deformation and fracture mechanisms caused by hydrogen observed in heat-treated steels similar to those caused by hydrogen and helium in tritium charged steels?

Purpose

In this study, the deformation and fracture toughness properties of heat treated stainless steels pre-charged with hydrogen were investigated and compared to the deformation and fracture toughness values of stainless steels pre-charged with tritium.

Material – Forged Type 21-6-9 Stainless Steel
Heat Treatments Used

- None
- 773 K: 0, 17, 1, 10, 24 Hours
- 873 K: 0, 17, 1, 10, 24 Hours
- Hydrogen Charging after Heat Treatments
- 623 K: 24 or 68 MPa for two weeks
- Tritium Charging and Aging No Prior Heat Treatment
- 623K: 24 MPa for two weeks
- Aging at 243 K for up to 5 years

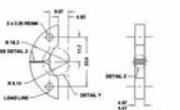
Experimental Procedure

Compositions of Stainless Steel Forgings (Weight %)

Material	Cr	Ni	Mn	Mo	C	Si	P	S	N	Co	O	Al
21-6-9 HERF	19.4	6.4	8.5	-	.04	.33	.021	<.001	.28	-	.0022	<.001
21-6-9 CF Heat 1	19.1	6.7	9.9	-	.03	.41	.01	.004	.28	-	.001	.005
21-6-9 CF Heat 2	19.3	6.7	9.9	-	.03	.38	.01	.001	.28	-	.002	.004

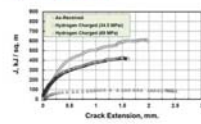
Notes:

CF – Conventionally Forged
HERF – High-energy-rate Forged

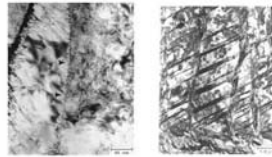
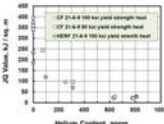


Fracture toughness specimens were cut from Type 21-6-9 forging and heat treated at 773 K or 873 K for times up to 24 h. After heat treatment, about one-half of the samples were pre-charged with hydrogen gas at 34.6 MPa and 623 K for 2 weeks. Fracture toughness testing was conducted on the non-charged and pre-charged samples and compared to earlier measurements on the same steels pre-charged with tritium gas.

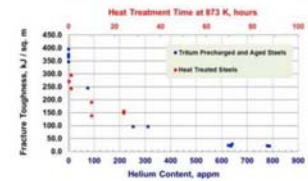
Results



Hydrogen, Tritium, and Helium Induced Fracture Toughness Reductions

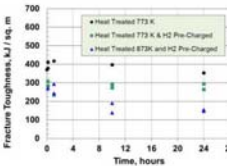


Helium Effects on Fracture and Deformation



Hydrogen-Precharged Steels That Are Given a Prior Heat Treatments at 873 K Show Similar Fracture Toughness Reductions and Fracture Modes As Tritium-Precharged Samples

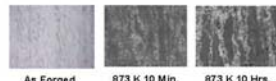
Prior work showed a reduction in fracture toughness properties after hydrogen and tritium pre-charging. Further reductions in fracture toughness occur over time in tritium pre-charged steels due to the build-in of helium from tritium decay. Tritium decay results in the nucleation of nanometer sized helium bubbles within the steel microstructure which enhances tritium embrittlement by hardening the microstructure and promoting deformation by twinning.



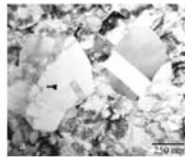
Hydrogen Precharged Stainless Steels Exhibit a Reduction in Fracture Toughness Properties after a Prior Heat Treatment in the Sensitization Range.

Uncharged specimens heat treated at 773 K showed no effect on toughness as a function of time at heat treatment temperature. These specimens exhibited no observable differences in grain structure or carbide distribution through the range of heat treatment times. Specimens heat treated at 773 K and hydrogen-charged showed a general reduction in fracture toughness of about 30% at all heat treatment times when compared to the uncharged.

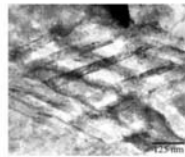
Hydrogen-charged and tested samples heat treated at 873 K had fracture toughness values that dropped off with increasing time at temperature to about 50% of the initial values.



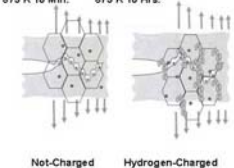
As Forged 873 K 10 Min. 873 K 10 Hrs.



Heat Treatment Resulted in the Formation of Patches of Recrystallized Grains in the Forged Matrix



Heavy Deformation Twinning on Multiple Slip Planes in Hydrogen-Charged Samples.



In hydrogen-charged samples, deformation by twinning is concentrated in the unrecrystallized grains. In tritium-charged samples concentrated deformation by twinning is brought on by the hardening effect of decay helium bubbles.

A microstructural analysis of foils from near the fracture surface of the heat treated steels showed that the fracture toughness reduction observed at the longer heat treatment times was not caused by an increase in the number of grain boundary carbides (which was the expected mechanism) but due to an increase in patches of recrystallized grains in the forged matrix. The increased number of recrystallized grains enhanced the deleterious effect of hydrogen on toughness by increasing the amount of deformation twinning. Heavy deformation twinning on multiple slip planes was typical for the hydrogen-charged samples; whereas, in the non-charged samples, less twinning was observed and was generally limited to one slip plane. In this image two different variants of deformation twins have formed in an unrecrystallized grain. Furthermore, deformation (via twinning) was not concentrated in the recrystallized grains; twinning was more prevalent in the unrecrystallized grains in the TEM specimens examined.

Summary and Conclusions

- Forged stainless steels pre-charged with tritium gas exhibit an aging effect: Fracture toughness values decrease with aging time due to an increasing concentration of helium-3 within the microstructure from tritium decay.
- Forged stainless steels pre-charged with hydrogen gas that are given a prior heat treatment also exhibit an aging effect: Fracture toughness values decrease with increasing heat treatment time in the temperature range 773 K to 873 K.
- The fracture toughness reduction observed in heat-treated steels was caused by an increase in the number of recrystallized grains in the forged matrix. The patches of recrystallized grains enhanced the deleterious effect of hydrogen on toughness by concentrating deformation and increasing the amount of deformation twinning in the unrecrystallized matrix.
- The fracture and deformation mechanisms caused by hydrogen in heat-treated steels appear to be similar to the fracture and deformation mechanisms caused by tritium in steels with decay helium.

We Put Science To WorkSM

SRNL is managed and operated for the U.S. Department of Energy by Savannah River Nuclear Solutions, LLC

*This manuscript has been authored by Savannah River Nuclear Solutions, LLC under Contract No. DE-AC05-00OR21400 with the U.S. Department of Energy. The United States Government retains, and the publisher, by accepting the article for publication, acknowledges that the United States Government retains a non-exclusive, paid-up, irrevocable, exclusive license to publish or reproduce the published form of the work, or allow others to do so, for the United States Government purposes.

Figure 11. 2012 International Hydrogen Conference Presentation on “The Effects of Hydrogen and Tritium and Heat Treatment on the Deformation and Fracture Toughness Properties of Stainless Steels.

VIII. SUMMARY

New experimental programs designed to measure the effects of the forging process and materials on tritium compatibility are underway. The programs are designed to investigate the effects of hydrogen, tritium, and decay helium on the fracture toughness properties of reservoir forgings. These programs will be unique because the effects of tritium on actual reservoir forgings have not been measured until now. Samples have been cut from a variety of forgings and exposed to hydrogen or tritium gas. Samples will be aged for up to five years and fracture mechanics tests conducted to measure the effect of decay helium on properties. Tests will be conducted in air until facilities are available for testing in high-pressure hydrogen gas. The specific objectives of each program are:

1. For Type 316L Cup forgings, measure the effect of crack orientation and in-part fracture toughness property variation.
2. For Type 304L Block forgings, measure fracture toughness properties for low-yield strength and high-yield strength forgings.
3. For Type 21-6-9 Brick forgings, measure effect of crack orientation and in-part fracture toughness variability. Also investigate sample size and geometry effect of fracture toughness and compare results with measurements conducted at LANL, SNL, and AWE.
4. For Type 304L forgings, measure effect of forging process, forging temperature, and prior anneal on fracture toughness properties. Also, measure forging strain rate effects on toughness and tritium compatibility by measuring toughness for forgings made using HERF, Mechanical Press, Screw Press, and Hydraulic Press.
5. For Type 21-6-9 Brick and Type 304L Block forgings, measure sustained load cracking thresholds using bolt-loaded specimens held in high pressure hydrogen environments.

IX. REFERENCES

1. Michael J. Morgan and Glenn K. Chapman, "Forging Effects on Fracture Toughness Properties of Tritium-Charged-and-Aged Stainless Steels – Program Plan and Initial Results", SRNL-TR-2011-00321, November 2011.
2. SRNL-L4400-2012-00027, SRNL Correspondence from M.J. Morgan to T.S. McGee, 10-9-2012, "SRNL Data Sheet Tritium Charging Conditions Run 2012-1".
3. SRNL-L4400-2012-00031, SRNL Correspondence from M.J. Morgan to G.D. Levi, 11-8-2012, "SRNL Data Sheet Tritium Charging Conditions Run 2012-2"

4. M. J. Morgan and G. K. Chapman, "Cracking Thresholds and Fracture Toughness Properties of Tritium-Charged-and-Aged Stainless Steels", WSRC-TR-2010-00393, Savannah River National Laboratory, Washington Savannah River Company, Savannah River Site, Aiken, SC, December, 2010.
5. M. J. Morgan and D. Lohmeier, "Threshold Stress Intensities and Crack Growth Rates In Tritium-Exposed HERF Stainless Steels", Hydrogen Effects on Material Behavior, N. R. Moody and A. W. Thompson, eds., pp. 459-468, TMS, Warrendale, PA (1990).
6. M. J. Morgan and M. H. Tosten, "Tritium and Decay Helium Effects on Cracking Thresholds and Velocities in Stainless Steels", Fusion Technology, Vol. 39, pages 590-595, 2001.
7. ASTM E 1681-03 "Standard Test Method for Determining Threshold Stress Intensity Factor for Environmental Assisted Cracking of Metallic Materials", 2008 Annual Book of ASTM Standard Volume 3.01 Metals-Mechanical Testing; Elevated and Low-Temperature Tests; Metallography, American Society for Testing and Materials, 2008.
8. Conceptual Design Package, Project Y-642, Enhanced Fracture Toughness Tester, 774-A, M-CDP-A-00014 Rev.0, prepared by William P. Lenartz, 9-21-2011.
9. SRNL-L4000-2012-00021, SRNL Correspondence from M.J. Morgan to J.M. Shappell, "Tritium Content in Samples and Charging Assembly – 1st Tritium Charging Run", 11-1-2012.
10. SRNL-L4400-2012-00032, SRNL Correspondence from M.J. Morgan to J. M. Shappell, "Tritium Content in Samples and Charging Assembly – 2nd Tritium Charging Run", 11-27-2012.
11. K. E. Kain, "Finite-Difference Program for Hydrogen Diffusion", DP-1738 March, 1987.
12. Calculation S-CLC-A-00147 "Tritium Content and Off-Gassing Rates for Samples Used in the Enhanced Fracture Toughness Tester" by Michael Morgan.
13. M. J. Morgan, "Cracking Thresholds and Fracture Toughness Properties of Tritium-Charged-and-Aged Stainless Steels", Submitted for Publication in the *Proceedings of the 2012 International Hydrogen Conference*, ASME, September 9-12, Moran, Wyoming.
14. M. J. Morgan, "Cracking Thresholds and Fracture Toughness Properties of Tritium-Charged-and-Aged Stainless Steels", SRNL-STI-2012-12-00531.

15. M. J. Morgan, M.H. Tosten, and G. K. Chapman, “The Effects of Hydrogen, Tritium, and Heat Treatment on the Deformation and Fracture Toughness Properties of Stainless Steel”, Submitted for Publication in the *Proceedings of the 2012 International Hydrogen*, ASME, September 9-12, Moran, Wyoming.
16. M. J. Morgan, M.H. Tosten, and G. K. Chapman, “The Effects of Hydrogen, Tritium, and Heat Treatment on the Deformation and Fracture Toughness Properties of Stainless Steel”, SRNL-STI-2012-12-00532.